# **On the Anharmonic Contribution to the Specific Heat of Monatomic Face-Centered Cubic Crystals**

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We have presented a method for an exact calculation of the two lowest-order, cubic  $(F_3)$  and quartic  $(F_4)$ , perturbation terms in the Helmholtz free energy  $(F)$ of an anharmonic crystal in the high temperature limit. The method is applicable to a nearest-neighbor central force model of a monatomic fcc crystal for any phenomenological two-body potential  $\phi(r)$ . The calculation of  $F_3$  and  $F_4$  requires a knowledge of the six dimensionless Brillouin zone (BZ) sums, as a function of a parameter  $a_1$  depending on the first and second derivatives of  $\phi(r)$ . These sums are calculated to a high degree of accuracy for a mesh size of 308,000 points in the whole BZ in the range  $-0.1 \le a_1 \le +0.1$  in steps of 0.02. The linear temperature dependent anharmonic contribution to the specific heat at constant volume, calculated for the elements Pb, Ag, Ni, Cu, A1, Ca, and Sr, from the Morse and Rydberg potentials, is found to be positive in all cases, with the exception of Pb. In this case the Morse potential gives a negative sign. The predictions of theory are in agreement with experiments where the data is available (e.g., Cu, AI, and Pb).

KEY WORDS: Helmholtz free energy; anharmonic fcc crystal; specific heat; perturbation theory; Brillouin zone sums.

## 1. INTRODUCTION

During the past 20 years, many calculations have been performed of the two lowest-order anharmonic free energy  $(F_3$  and  $F_4$ ) terms of rare gas anharmonic crystals. For an exhaustive review of the calculations of  $F_3$  and  $F_4$  for rare gas solids, see [1].  $F_3$  and  $F_4$  arise from the cubic and quartic terms in the Taylor series expansion of the crystal potential energy. In the high tempera-

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ture limit  $(T > \theta_p)$ , where  $\theta_p$  is the Debye temperature,  $F_3$  and  $F_4$  both contribute to  $C<sub>v</sub>$  (specific heat at constant volume), a term linear in temperature  $(T)$ , i.e.,

$$
C_{\nu}(\text{anharmonic}) = C_{\nu}^{A} = 3R \left\{ -\frac{2}{3} \left[ a_{3}(\nu) + a_{4}(\nu) \right] \right\} K_{B} T \qquad (1)
$$

where the volume dependent coefficients  $a_3(v)$  and  $a_4(v)$  in Eq. (1) are obtained from  $F_3$  and  $F_4$ , respectively,  $K_B$  is the Boltzmann constant, and R is the gas constant.

One of the earlier calculations of  $a_3(v)$  and  $a_4(v)$  was carried out in the high temperature (HT) limit by Stern [2] for a special nearest-neighbor model of bcc Na crystal. The next exhaustive calculations of  $a_3(v)$  and  $a_4(v)$ in the HT limit were performed by Maradudin et al. [3] for a nearestneighbor central force model of a fcc crystal in the leading-term approximation. The leading-term approximation involves retaining only the highestorder radial derivative in each of the expressions for the cartesian tensor derivatives of the two-body potential  $\phi(r)$ , which occurs in the mathematical expressions for  $F_3$  and  $F_4$ . Maradudin et al. [3] also corrected an error in the  $a_3(v)$  calculation by Stern [2] and performed their calculations at  $v = v_0$ , corresponding to the volume  $(v_0)$  at the minimum in  $\phi(r)$  at the nearest neighbor distance  $(r_0)$ , i.e.,  $\phi'(r_0) = 0$ .

In this paper, we present a calculation of  $F_3$  and  $F_4$  for the monatomic fcc structure, using the central force model potential,  $\phi(r)$ , without making the leading term approximation. The error [4] introduced in  $F_3$  and  $F_4$  in this approximation is 0.3% and 40%, respectively. Here we do not assume  $\phi'(r_0)$  = 0. Thus our calculations of  $F_3$  and  $F_4$  have been carried out for a set of values of the ratio  $a_1 = \phi'/[r(\phi'' - \phi'/r)]$  [5], where  $\phi'$  and  $\phi''$  are the first and second radial derivatives of  $\phi(r)$ . This ratio serves to characterize the various nearest-neighbor potentials  $\phi(r)$ , and, in addition, for a given potential  $\phi(r)$ , its variation can represent changes in the lattice constant  $a_0$  and hence the variation of  $F_3$  and  $F_4$  with volume. In other words, we obtian  $a_3(v)$  and  $a_4(v)$ at any volume for a given potential  $\phi(r)$ .

Often the parameters of the phenomenological potentials such as the Morse and Rydberg potentials of Girifalco and Weizer [6] and Varshni and Bloore [7], respectively, are determined by considering "all" neighbor interactions. For these potentials the ratio  $a_1$  is negative. Most cases are taken care of if we restrict  $a_1$  to the range  $-0.1 \le a_1 + 0.1$ . The calculations of  $F_3$  and  $F_4$ presented in this paper are in this range of  $a_1$  in steps of 0.02. The range of  $a_1$ in [5] was  $-0.02 \le a_1 \le 0.1$ , which is not adequate for the above potentials.

In Sec. 2, we present a summary of the expressions for  $F_3$  and  $F_4$  in terms of dimensionless sums. These dimensionless sums are functions of the frequency  $\omega(\mathbf{q} \, j)$  for a wave vector **q** and branch index *j.*  $\omega(\mathbf{q} \, j)$  are obtained by diagonalizing the dynamical matrix  $D_{\alpha\beta}(\mathbf{q})$ , which depends on the ratio  $a_1$ . Hence the dimensionless sums also depend on the ratio  $a_1$ .

The calculation of these dimensionless sums as a function of  $a_1$  is presented in Sec. 3. As an example, calculations of  $F_3$  and  $F_4$  and  $C_v^A$  for a number of materials are presented in Sec. 4 where, given any  $\phi(r)$  such as Morse or Rydberg potentials, it is shown how to find the value of the dimensionless sums required in the calculation of  $F_3$  and  $F_4$ .

A discussion of the numerical results obtained in this paper for a number of materials (Pb, Ag, Ni, Cu, A1, Ca, Sr) and the validity of the nearestneighbor model for metals are presented in Sec. 5, where it is concluded that the model used in this paper has its usefulness in metals involving large ion cores, such as Cu.

# **2. EXPRESSIONS FOR**  $F_3$  **AND**  $F_4$  **AND THE RATIO**  $a_1$

A complete derivation of the expressions for  $F_3$  and  $F_4$  in computational form for the "all" neighbor interaction potential  $\phi(r)$ , without making any approximation, for a monatomic fcc or bcc structure can be found in Shukla and Taylor [8]. Hence there is little point in repeating this derivation here. For our purposes we can write the expression for  $F<sub>4</sub>$ , for a nearest-neighbor central force model of an fcc crystal [4], in the following form:

$$
F_4 = \frac{N(K_B T)^2}{64 |B(r_0)|^2} \left[ D(r_0) S4A + \frac{2C(r_0)}{r_0} S4B + \frac{4B(r_0)}{r_0^2} S4C \right] \tag{2}
$$

where N is the Avogadro number; *S4A*, *S4B*, and *S4C* are the dimensionless single whole Brillouin zone sums;  $K_B$  and T have been defined in Eq. (1); and  $B(r_0)$ ,  $C(r_0)$ , and  $D(r_0)$  are the various combinations of the first four derivatives of  $\phi(r)$  evaluated at  $r_0$ , and their expressions in terms of these derivatives are given by

$$
B(r_0) = \phi''(r_0) - \frac{1}{r_0} \phi'(r_0)
$$
\n(3)

$$
C(r_0) = \phi'''(r_0) - \frac{3}{r_0} \phi''(r_0) + \frac{3}{r_0^2} \phi'(r_0)
$$
\n(4)

$$
D(r_0) = \phi^{\text{iv}}(r_0) - \frac{6}{r_0} \phi^{\prime\prime\prime}(r_0) + \frac{15}{r_0^2} \phi^{\prime\prime}(r_0) - \frac{15}{r_0^3} \phi^{\prime}(r_0)
$$
 (5)

where  $r_0$  is the nearest-neighbor distance  $(r_n)$ .

The expression for  $F_3$  is more complicated than  $F_4$ . For a nearestneighbor central force model of an fcc crystal, the complete expression for  $F_3$ can be written in the following form [4]:

$$
F_3 = -\frac{N(K_B T)^2 [C(r_0)]^2}{(96)(32)[B(r_0)]^3} [S3A + 6(RBC)S3B + (RBC)^2 S3C]
$$
 (6)

where *S3A, S3B,* and *\$3C* are dimensionless double Brillouin zone sums.  $RBC = 2B(r_0)/r_0C(r_0)$ , and the other symbols have been defined earlier in connection with Eq. (2).

The expressions for the dimensionless sums *S4A, S4B,* and *\$4C* arising in  $F_4$  are given in full in Ref. [8]. We will not reproduce them here, but, in order to see how they depend on the ratio  $a<sub>1</sub>$ , let us consider the full expression for *S4A* for a nearest-neighbor fcc crystal [3,8]:

$$
S4A = \frac{12}{N^2} \left[ \sum_{\mathbf{q}j} \frac{\left[ e_x(\mathbf{q}j) + e_y(\mathbf{q}j) \right]^2}{\lambda^2(\mathbf{q}j)} \left[ 1 - \cos \frac{a_0}{2} (q_x + q_y) \right] \right]^2 \tag{7}
$$

where the summations over the wave vector  $q$  and j are over the whole first Brillouin zone and branch index, respectively;  $e_{\alpha}(\mathbf{q}t)$  are the eigenvector components ( $\alpha = x, y, z$ ), and  $\lambda(q, j)$  are the dimensionless frequencies to be determined from the dynamical matrix  $D_{\alpha\beta}(\mathbf{q})$ . For a potential  $\phi(r)$  and the nearest-neighbor fcc crystal, the diagonal,  $D_{xx}(q)$  and the off-diagonal,  $D_{xy}(\mathbf{q})$ , matrix elements are given by [9]

$$
MD_{xx}(\mathbf{q}) = 2B(r_0)[2 - C_x(C_y + C_z) + 2a_1(3 - C_xC_y - C_yC_z - C_zC_x)]
$$
 (8)

$$
MD_{xy}(\mathbf{q}) = 2B(r_0)S_xS_y \tag{9}
$$

where, in Eqs. (8) and (9),  $C_i = \cos a_0 q_i$ ,  $S_i = \sin a_0 q_i (i = x, y, z)$ ; M is the atomic mass; and  $B(r_0)$  and the ratio  $a_1$  have been defined above.

The eigenvalue equation is given by

$$
\sum_{\beta} D_{\alpha\beta}(\mathbf{q}) e_{\beta}(\mathbf{q} j) = \omega^2(\mathbf{q} j) e_{\alpha}(\mathbf{q} j)
$$
 (10)

where  $\alpha$ ,  $\beta = x, \gamma, z$ , and  $j = 1,2,3$ .

The dimensionless frequencies  $\lambda(qj)$  in Eq. (7) are obtained from the scaling procedure

$$
\omega^2(\mathbf{q} j) = \frac{2B(r_0)}{M} \lambda^2(\mathbf{q} j) \tag{11}
$$

Clearly then,  $\lambda(qj)$  depends on the ratio  $a_1$ , and thus *S4A* depends on  $a_1$ . Similarly, the other dimensionless sums  $S4B$  and  $S4C$  in  $F_4$  depend on  $a_1$ .

In the same manner one can see that the other sums *S3A, S3B,* and *\$3C,*  involving double Brillouin zone summations and triple branch index summations in the expression for  $F_3$ , Eq. (6), also depend on  $a_1$ . The expressions for *S3A, etc., are lengthy and we do not reproduce them here. They can be* obtained from Refs. [3, 4, 8]

### 3. CALCULATION OF THE DIMENSIONLESS SUMS

The dimensionless sums *S4A, S4B, \$4C,* and *S3A, S3B, \$3C* can be calculated by the method described in Shukla and Wilk [4]. In Ref. [4], these sums were calculated for  $a_1 = 0$ . The calculation requires the knowledge of the following tensorial Brillouin zone sums for a direct lattice vector  $\mathbf{r}'$ :

$$
S_{xx}(l) = \sum_{qj} \frac{e_x(qj)e_x(qj)}{\lambda^2(qj)} \cos(q \cdot r^l)
$$
 (12)

$$
S_{xy}(l) = \sum_{q} \frac{e_x(qj)e_y(qj)}{\lambda^2(qj)} \cos(q \cdot r^l)
$$
 (13)

The other two-diagonal and off-diagonal tensors, viz.,  $S_{yy}(l)$ ,  $S_{zz}(l)$ ,  $S_{xz}$ (1), and  $S_{yz}$ (1) are obtained from the above by proper interchange of the indices. It has been shown in Ref. [4] how to reduce the whole Brillouin sums in Eqs. (12) and (13) to the  $1/48$ th portion of the zone and eliminate the j summation, using the following simple version of Born's theorem [ 10]:

$$
\sum_{i} \frac{e_{\alpha}(\mathbf{q} j)e_{\beta}(\mathbf{q} j)}{\lambda^2(\mathbf{q} j)} = [H^{-1}(\mathbf{q})]_{\alpha\beta}
$$

where  $[H^{-1}(q)]_{\alpha\beta}$  is the  $\alpha\beta$  element of the inverse dynamical matrix defined by Eqs. (8) and (9) with  $2B(r_0)/M = 1$ .

Our procedure in computing *S4A, S4B, \$4C,* and *S3A, S3B, \$3C* was to first calculate and store  $S_{\text{a}\beta}(I)$  for a sufficient number of direct lattice vectors **i.** This was done for 11 values of  $a_1$  in the range  $-0.1 \le a_1 \le +0.1$  in steps of 0.02.  $S_{a\beta}(I)$  are then obtained from Eqs. (12) and (13). In the actual calculation of *S4A, S4B, \$4C,* and *S3A, S3B, \$3C,* the appropriate sum was selected as needed by comparing the direct lattice vector in the argument of  $S_{\alpha\beta}(I)$  with those of the stored general set of direct lattice vectors.

In all the calculations, we have used a simple cubic mesh of wave vectors with  $q = \pi p / La_0$  and boundaries of the 1/48th portion of the first Brillouin

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a <sub>1</sub>	S4A	S4B	S4C
0.10	6.6424	36.1406	32.8783
0.08	7.3575	40.1996	36.9149
0.06	8.2049	45.0390	41.7913
0.04	9.2219	50.8877	47.7716
0.02	10.4613	58.0712	55.2378
0.00	12.0000	67.0682	64.7629
$-0.02$	13.9544	78.6124	77.2455
$-0.04$	16.5105	93.8898	94.1752
$-0.06$	19.9855	114.9579	118.2144
$-0.08$	24.9771	145.7607	154.6527
$-0.10$	32.7935	195.1482	215.9211

**Table I.** Dimensionless Sums,  $S4A$ ,  $S4B$ , and  $S4C$ , for Different Values of  $a<sub>1</sub>$ 

zone defined by  $L \ge p_x \ge p_y \ge p_z \ge 0$ ,  $p_x + p_y + p_z \le 1.5L$ , where  $p_x, p_y, p_z$  and L are integers and  $a_0$  is the lattice constant. This yields  $4L^3$  points in the whole zone. In Ref. [4], the step length  $L = 7$ , corresponding to 1372 points in the whole zone, was used in computing *S4A, S4B, \$4C* and *S3A, S3B, \$3C* for  $a_1 = 0$ . Here, in this paper, we have calculated these sums for  $L = 30$ , which corresponds to 308,000 points in the whole zone. We have presented in Tables I and II the sums *S4A, S4B, \$4C* and *S3A, S3B, \$3C,* respectively, for 11 values of  $a_1$  in the range  $-0.1 \le a_1 \le +0.1$  in steps of 0.02.

# **4. CALCULATION OF**  $F_3$ **,**  $F_4$  **AND**  $C_1^A$

The cubic and quartic terms of the Helmholtz freen energy,  $F_3$  and  $F_4$ , can be calculated from Eqs.  $(2)$ - $(6)$  presented in Sec. 3 for any two-body

a <sub>1</sub>	S3A	S3B	S3C
0.10	70.5913	38.4839	305.3829
0.08	82.3512	44.7611	356.0845
0.06	97.0710	52.5874	419.5128
0.04	115.8116	62.5078	500.2258
0.02	140.1570	75.3319	605.0335
0.00	172.5623	92.3066	744.5057
$-0.02$	217.0117	115.4410	935.8389
$-0.04$	280.3586	148.1617	1208.742
$-0.06$	375.3332	196.7660	1618.819
$-0.08$	528.2631	274.0925	2282.677
$-0.10$	803.4675	410.8234	3493.464

**Table II.** Dimensionless Sums, S3A, S3B, and S3C, for Different Values of  $a_1$ 

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potential  $\phi(r)$  in the following manner. First we find the derivatives  $\phi'(r)$  and  $\phi''(r)$  of  $\phi(r)$  and then compute the ratio  $a_1$ , defined by

$$
a_1 = \frac{(\phi'/r)}{(\phi'' - \phi'/r)}\bigg|_{r=r_0}
$$
 (14)

For this value of  $a_1$ , the values of the dimensionless sums *S4A*, *S4B*, *S4C* and *S3A, S3B, S3C, which are functions of*  $a<sub>1</sub>$ *, are obtained by interpolation from* the Tables I and II, respectively. Next, we obtain the other derivatives of  $\phi(r)$ and compute  $B(r_0)$ ,  $C(r_0)$  and  $D(r_0)$  from Eqs. (3)-(5). Finally, with all this,  $F_4$  and  $F_3$  are easily computed from Eqs. (2) and (6), respectively. The anharmonic contribution to  $C_v$  from  $F_3$  and  $F_4$  is obtained from the following:

$$
C_v^A = -T \frac{\partial^2}{\partial T^2} (F_3 + F_4) = 3RAK_B T = 3RBT, \text{ where } B = AK_B \text{ (15)}
$$

As an example of the above description of finding  $F_3$ ,  $F_4$ , and  $C_v^A$  for a given  $\phi(r)$  from Tables I and II, we consider the following two widely used phenomenological potentials: (a) the Morse potential,  $\phi_M(r)$ , and (b) the Rydberg potential,  $\phi_R(r)$ , defined by

$$
\phi_M(r) = D[\exp\{-2\alpha(r-\gamma_0)\} - 2\exp\{-\alpha(r-\gamma_0)\}] \qquad (16)
$$

$$
\phi_R(r) = -D[1 + \alpha(r - \gamma_0)] \exp[-\alpha(r - \gamma_0)] \tag{17}
$$

where the potential parameters D,  $\alpha$ , and  $\gamma_0$ , appearing in  $\phi_M(r)$  and  $\phi_R(r)$ , have been determined for a number of monatomic fcc materials by Girifalco and Weizer [6] and Varshni and Bloore [7], respectively.

Following the procedure outlined above, we have calculated numerical values of  $a_1, F_3, F_4, F = F_3 + F_4$ , and the anharmonic coefficient B in Eq. (15) for the elements Pb, Ag, Ni, Cu, Al, Ca, and Sr. These results for  $\phi_M(r)$  and  $\phi_R(r)$  potentials [6, 7] are presented in Tables III and IV, respectively.

### 5. DISCUSSION

The method presented in this paper of an exact calculation of the two lowest-order perturbation terms in the Helmholtz free energy  $(F)$  of an anharmonic crystal, viz.,  $F_3$  and  $F_4$ , is applicable to a nearest-neighbor central force model of a monatomic fcc crystal in the high temperature limit. Given any phenomenological two-body potential  $\phi(r)$ ,  $F_3$  and  $F_4$  and their contribution to  $C_r^A$  can be calculated from the equations given in the previous sections. We performed these calculations for seven elements (Pb, Ag, Ni, Cu, A1, Ca,

<b>Elements</b>	a,	$F_{\rm 3}$	$F_{4}$	$F = F_1 + F_4$	B
Pb	$-0.048$	$-0.34940$	0.37142	0.02202	$-2.027$
Ag	$-0.052$	$-0.24382$	0.23984	$-0.00399$	0.367
Ni	$-0.067$	$-0.17853$	0.13487	$-0.04367$	4.019
Сu	$-0.069$	$-0.21816$	0.15615	$-0.06201$	5.707
Al	$-0.076$	$-0.27377$	0.17132	$-0.10245$	9.429
Ca	$-0.083$	$-0.45473$	0.23819	$-0.21654$	19.929
Sг	$-0.084$	$-0.48558$	0.25233	$-0.23325$	21.467

**Table III.** Cubic,  $F_3$ , quartic,  $F_4$ , the total free energy, F, and the coefficient, B, in  $C_v^A/3R = BT$ , for the Morse Potential<sup>a</sup>

<sup>a</sup>F<sub>3</sub>, F<sub>4</sub>, and F are in units of  $10^{12} N(K_B T)^2$  erg<sup>-1</sup> and B is in units of  $10^{-6}/K$ .

and Sr), employing the Morse and Rydberg phenomenological potentials. In all our calculations we have used the Morse and Rydberg potential parameters determined by Girifalco and Weizer [6], and Varshni and Bloore [7], respectively. The value of the parameter  $b$  quoted in Table II of Ref. [7] for Ca is in error. The correct value communicated to the author by Y. P. Varshni is  $b = 1.1884$ . This value has been used in the present calculations of  $F_3$ ,  $F_4$ , and B for Ca.

It is interesting to note that both the potentials predict a positive anharmonic contribution for  $C<sub>v</sub>$  in six elements, viz., Ag, Ni, Cu, Al, Ca, and Sr, but for Pb the signs are opposite.

There are two sets of predictions of  $C_v^A$  from the experimental data on the heat capacity in Al  $[11, 12]$ . The prediction of  $C_v^A$  being positive from both the potentials agrees with Brooks and Bingham [11]. Leadbetter [12] predicts a negative  $C_{v}^{A}$ . Currently we are analyzing this situation in Al, and the results of our findings will be reported later. For Pb the two potentials give opposite results: positive  $C_v^A$  from Rydberg and negative  $C_v^A$  from Morse. The

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Elements	$a_1$	$F_{\rm 3}$	$F_{4}$	$F - F_1 + F_4$	B
Pb	$-0.042$	$-0.32612$	0.31132	$-0.01479$	1.362
Ag	$-0.049$	$-0.22106$	0.18480	$-0.03626$	3.337
Ni	$-0.062$	$-0.16057$	0.10514	$-0.05543$	5.101
Cu	$-0.064$	$-0.19458$	0.11932	$-0.07525$	6.926
Al	$-0.071$	$-0.24209$	0.12706	$-0.11503$	10.587
Ca	$-0.078$	$-0.39899$	0.17110	$-0.22789$	20.974
Sг	$-0.078$	$-0.42799$	0.18358	$-0.24440$	22.494

**Table IV.** Cubic,  $F_3$ , quartic,  $F_4$ , the total free energy, F, and the coefficient, B, in  $C_v^A = BT$ , for the Rydberg Potential<sup>a</sup>

 ${}^4F_3$ ,  $F_4$ , and F are in units of  $10^{12} N(K_B T)^2$  erg<sup>-1</sup> and B is in units of  $10^{-6}/K$ .

latter prediction is in agreement with the analysis (experimental) of Leadbetter [12]. For Cu the prediction of the sign of  $C_v^A$  from both the potentials agrees with that of Brooks' [13] analysis of the experimental heat capacity data.

Now the question arises: Is the nearest-neighbor model useful in the calculations of the above type? It would seem that for metallic crystals the nearest-neighbor central force (nncf) model is inadequate if the volume dependent terms arising from the electron gas effects are not included in the potential function. As a matter of fact, the interatomic forces, even in such simple metals as Na, K, etc., are long range and oscillatory and, in addition, they change as the crystal volume changes. The calculation of  $F_3$  and  $F_4$  in such cases is very time consuming, with the additional difficulty of convergence of the sums arising in  $F_4$ . A full description of the method of  $F_3$  and  $F_4$ calculation and a discussion of the convergence difficulties can be found in the paper of Shukla and Taylor [8]. In such cases the nncf model has very little or no usefulness.

On the other hand, in another report [14] on the anharmonic calculations in metals such as copper, where the interatomic forces are known to exist primarily between nearest-neighbor atoms, due to the overlap of the  $3d$ orbitals on adjacent atoms (overlap energy), the nncf model was found to be useful. The nncf model is to some extent justified in metals like Cu as seen by a direct calculation [15, 16] of the electron-ion contribution to the forces, which indicates that the convergence is rapid, and by the fit to experimental phonon dispersion curves given by a nearest-neighbor model [17]. In the nncf model of Cu employed by Cowley and Shukla [14], the overlap energy was represented by a Born-Mayer term of the type

$$
V_{BM}(r) = \alpha \exp\left[\rho(\gamma_0 - r)/\gamma_0\right]
$$
 (18)

where  $\alpha$ ,  $\rho$ , and  $\gamma_0$  are potential parameters. The total energy was taken as  $V_{BM}(r)$  plus the volume dependent electron gas contributions describing the kinetic, correlation, and exchange energies. It was shown in this work that the volume dependent electronic energy terms can be included in the potential function, acting between nearest neighbors, if the parameter  $\alpha$  in  $V_{BM}(r)$  is volume dependent. The method presented in this paper of the  $F_3$  and  $F_4$ calculation is applicable to the above type of potentials in Cu as well as other similar potentials in monatomic fcc structures.

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